

THE REACTIVITY OF TiO<sub>2</sub> WITH DIFFERENT  
PREPARATION HISTORIES IN SOLID-STATE REACTIONS  
IN CaSO<sub>4</sub>–TiO<sub>2</sub> SYSTEMS

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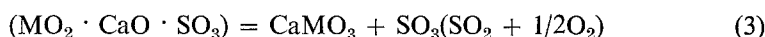
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Influences of the preparation histories of TiO<sub>2</sub> on its reactivity in solid-state reactions in CaSO<sub>4</sub>–TiO<sub>2</sub> systems were studied by means of TG, DTA and isothermal kinetic experiments.

In order to estimate the reactivities of solids, it was necessary to note the behaviors in the initial reaction step in the selected reaction system. It was concluded that the reactivity of TiO<sub>2</sub> was dependent on the structure difference due to the different preparation histories, and the greater the stability of the anatase form, up to higher temperature, the higher the reactivity of TiO<sub>2</sub>.

In a previous paper [1], the solid-state reactions in mixed systems of CaSO<sub>4</sub> with 12 metal oxides were investigated by means of gas-flow type DTA. In the CaSO<sub>4</sub>–[oxide]<sub>II</sub> systems, where [oxide]<sub>II</sub> can be any one of Al<sub>2</sub>O<sub>3</sub>, SiO<sub>2</sub>, TiO<sub>2</sub>, Cr<sub>2</sub>O<sub>3</sub>, MnO<sub>2</sub> or  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>, decomposition of CaSO<sub>4</sub> was promoted by the oxide. It was supposed that this promotion is connected with the formation of binary metal oxide through intermediates, which are CaSO<sub>4</sub> incorporated with an oxide, in the initial steps of the reaction:



where parentheses show the intermediates which suggest the distortion of the lattice, and [MO<sub>2</sub>]<sub>II</sub> shows [oxide]<sub>II</sub>. It seems reasonable to assume that step (3) determines the overall rates, because the rate of decomposition of the CaSO<sub>4</sub>–[oxide]<sub>II</sub> system was markedly affected by the atmospheres used.

The present paper is concerned with TG, DTA and isothermal studies on the influence of the preparation history of TiO<sub>2</sub> on its reactivity as regards the initial reaction step in TiO<sub>2</sub>–CaSO<sub>4</sub> systems.

### Experimental

#### *Materials*

Six TiO<sub>2</sub> samples and their preparation methods and physical properties are shown in Table 1. Sample S–2 was prepared by calcining a precipitate, obtained by the hydrolysis of 20 g TiCl<sub>4</sub> in 100 ml H<sub>2</sub>O, at 900° for 2 hr in air. S–3 was

Table 1  
 $\text{TiO}_2$  samples

Preparation methods			Physical properties		
samples	starting materials	calcining <sup>c</sup>	anatase contents, $F_A$ , wt. %	$\text{SO}_3$ contents, $C_{\text{SO}_3}$ , wt. %	BET surfaces, $S_1$ , m <sup>2</sup> /g
C-A	Commercial anatase <sup>a</sup>	700°, 2 hr	100	0.01	10.9
C-R <sub>1</sub>	C-A	1200°, 1 hr	0	—	—
C-R <sub>2</sub>	Commercial rutile <sup>b</sup>	700°, 1 hr	0	—	13.4
S-2	Hydrolyzed $\text{TiCl}_4$ , (S-2')	900°, 2 hr	0	0	1.4
S-3	Hydrolyzed $\text{TiCl}_4$ with $(\text{NH}_4)_2\text{SO}_4$ , (S-3')	700°, 2 hr	100	0.50	18.0
S-5	Hydrolyzed $\text{Ti}(\text{SO}_4)_2$ , (S-5')	700°, 2 hr	62.7	0.12	14.6

<sup>a</sup> Kanto Chemical Co.

<sup>b</sup> Commercial pigment.

<sup>c</sup> In air.

prepared by calcining a precipitate, obtained by the hydrolysis of 20 g  $\text{TiCl}_4$  with 5 g  $(\text{NH}_4)_2\text{SO}_4$  additive in 100 ml  $\text{H}_2\text{O}$ , at 700° for 2 hr in air. S-5 was prepared by calcining a precipitate, obtained by the hydrolysis of 19.5 g  $\text{Ti}(\text{SO}_4)_2$  in 100 ml  $\text{H}_2\text{O}$ , at 700° for 2 hr in air. All the hydrolysis reactions were carried out at 100°. Each precipitate (hydrated  $\text{TiO}_2$ , S-2', S-3' or S-5') was washed with warm water until the washings had no  $\text{Cl}^-$  or  $\text{SO}_4^{2-}$ , followed by filtration and drying at 100°.  $\text{TiCl}_4$ ,  $(\text{NH}_4)_2\text{SO}_4$  and  $\text{Ti}(\text{SO}_4)_2$  were GR reagents from Kanto Chemical Co., and were used without further purification. C-A, C-R<sub>1</sub> and C-R<sub>2</sub> were prepared by calcining commercial  $\text{TiO}_2$  under the conditions shown in Table 1.  $\text{CaSO}_4$  was prepared by calcining commercial  $\text{CaSO}_4 \cdot 2 \text{H}_2\text{O}$  (GR reagent from Kanto) at 550° for 1 hr in air.

Prior to the preparation of the mixture used for the solid-state reaction in the  $\text{TiO}_2$ - $\text{CaSO}_4$  system, the  $\text{TiO}_2$ - $\text{CaSO}_4$  system, the  $\text{TiO}_2$  and  $\text{CaSO}_4$  were ground to pass a 325 mesh sieve. Particle sizes in the range 0.3-1.0  $\mu\text{m}$  was measured under an electron-microscope, except for the C-R<sub>1</sub> sample, where it was about 2  $\mu\text{m}$ . Equimolar mixtures of  $\text{TiO}_2$  and  $\text{CaSO}_4$  were blended in an agate mortar for 45 min. Completeness of mixing was checked by measuring the amount of  $\text{Ca}^{2+}$  in several parts of the sample after mixing.

#### TG and DTA

The TG apparatus used for the reactivity tests of five  $\text{TiO}_2$  samples with different preparation histories consisted of a quartz spring (sensitivity 3.26 mg/mm) and a quartz reaction tube ( $d = 40$  mm) connected to a vacuum system. The sample of 0.1 g of a mixtures of  $\text{TiO}_2$  and  $\text{CaSO}_4$  was placed in a cylindrical quartz dish

(10 × 10 mm) and the weight decrease of the sample under 150 torr N<sub>2</sub> was measured with a cathetometer as a function of temperature. A temperature increasing rate of 5°/min was used. The net weight decrease was estimated from the difference between the data measured in the TiO<sub>2</sub>-CaSO<sub>4</sub> system and the baseline obtained for α-Al<sub>2</sub>O<sub>3</sub>. Temperature was measured with a CA thermocouple attached at the outer wall of the reaction tube.

The gas-flow type DTA apparatus [1] was used under the conditions of flowing N<sub>2</sub> (100 ml/min) and a heating rate of 5.8°/min for TiO<sub>2</sub> (C-A and C-R<sub>1</sub>)-CaSO<sub>4</sub> systems.

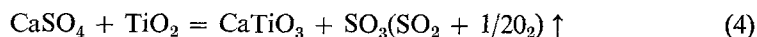
In order to characterize the preparation histories of the samples S-2, 3 and 5, the calcination processes of hydrated TiO<sub>2</sub>, S-2', 3' and 5', were investigated by means of TG and DTA. TG experiments were carried out with a Cahn electro-balance Model RG. A heating rate of 4.5°/min and a static atmosphere of 150 torr N<sub>2</sub> were used. DTA experiments were carried out with a Thermoflex 8001 apparatus (Rigaku Denki). A heating rate of 10°/min and static air atmosphere were used.

#### *Isothermal experiment*

The TG apparatus was also used for the isothermal study in TiO<sub>2</sub>-CaSO<sub>4</sub> systems in the temperature range 900 to 1000°. In order to start the reaction, the sample tube was inserted into the furnace heated up to the prescribed temperature, followed by the measurement of the weight decrease of the sample under 150 torr N<sub>2</sub> as a function of time. There was some uncertainty in the fractional conversion measured in an initial period of time, because about 15 minutes was required to attain the fixed reaction temperature after the sample was inserted into the heated furnace. Consequently, the data after 20 minutes were used for the kinetic considerations.

#### *Fractional conversion (α)*

The fractional decomposition of CaSO<sub>4</sub> was estimated from the weight decrease of the sample by assuming the reaction:



The fractional formation of CaTiO<sub>3</sub> was determined separately by chemical analysis. Chelatometry was used to determine the amount of Ca<sup>2+</sup> unreacted. CaTiO<sub>3</sub> was the only product, and no CaO was detected. The values obtained by TG were 1-2% higher than those by chemical analysis. However, this is within the experimental error in the accuracy of chemical analysis.

#### *Anatase content (F<sub>A</sub>), SO<sub>3</sub> content (C<sub>SO<sub>3</sub></sub>) and surface area (S)*

The F<sub>A</sub> (wt%) of the sample was estimated on the basis of the following equation [2].

$$F_A = \frac{1}{1 + 1.26I_R/I_A} \quad (5)$$

where  $I_A$  and  $I_R$  are the X-ray diffraction intensities at  $2\theta = 25.3^\circ$  for anatase and  $27.5^\circ$  for rutile, respectively. The  $C_{\text{SO}_3}$  (wt%) of the sample was determined by the chemical analysis of  $\text{SO}_3$  and/or  $\text{SO}_2$  liberated when the sample was calcined at  $1200^\circ$  in flowing  $\text{O}_2$ . The  $S$  ( $\text{m}^2/\text{g}$ ) of the sample was estimated by the BET method from the amount of  $\text{N}_2$  adsorbed at  $-195^\circ$ .

## Results and discussion

### *Changes of hydrated $\text{TiO}_2$ during calcining*

Figure 1 shows the DTA and TG curves for three hydrated  $\text{TiO}_2$  samples. An endothermic peak which corresponds to the dehydration appears at  $150$ – $200^\circ$  for each sample. An endothermic peak which corresponds to the decomposition

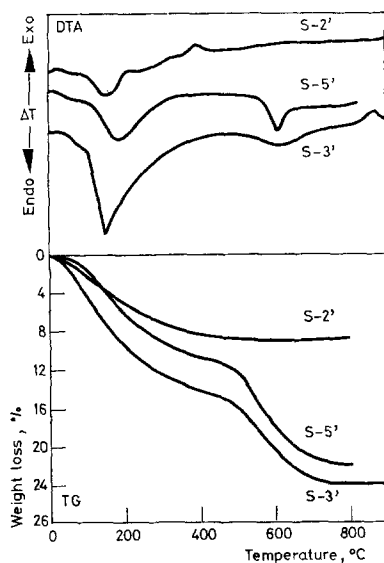


Fig. 1. DTA and TG curves for hydrated  $\text{TiO}_2$ . Heating rate:  $10^\circ/\text{min}$ ; atmosphere: static air

appears at about  $600^\circ$  for samples S-3' and 5'. The weight decreases shown in the TG curves correspond to these endothermic peaks. The TG curve of S-2' shows a weight decrease in only one step, which corresponds to the dehydration. Assuming that the weight decrease above  $500^\circ$  is due to the release of  $\text{SO}_3$  [3, 4], the compositions of S-2', 3' and 5' were evaluated to be  $\text{TiO}_2 \cdot 0.49 \text{H}_2\text{O}$ ,  $\text{TiO}_2 \cdot 0.82 \text{H}_2\text{O} \cdot 0.13 \text{SO}_3$  and  $\text{TiO}_2 \cdot 0.60 \text{H}_2\text{O} \cdot 0.15 \text{SO}_3$ , respectively. It is thought that the exothermic peaks at  $400^\circ$  for S-2' and at  $870^\circ$  for S-3' correspond to the crystallization of rutile and the transition from anatase to rutile, respectively.

Figure 2 shows the changes of  $F_A$ ,  $C_{\text{SO}_3}$  and  $S$  as functions of the calcination temperature. The calcination time of samples for  $F_A$  values is 1 hr, and for  $C_{\text{SO}_3}$  or  $S$  values is 2 hr. In  $S-3'$  and  $S-5'$ ,  $C_{\text{SO}_3}$  decreases with increasing calcination temperature. It is thought that the  $C_{\text{SO}_3}$  decrease results from the transition from anatase to rutile, or the decrease of  $F_A$  and  $S$ . The inflection points of the  $F_A$

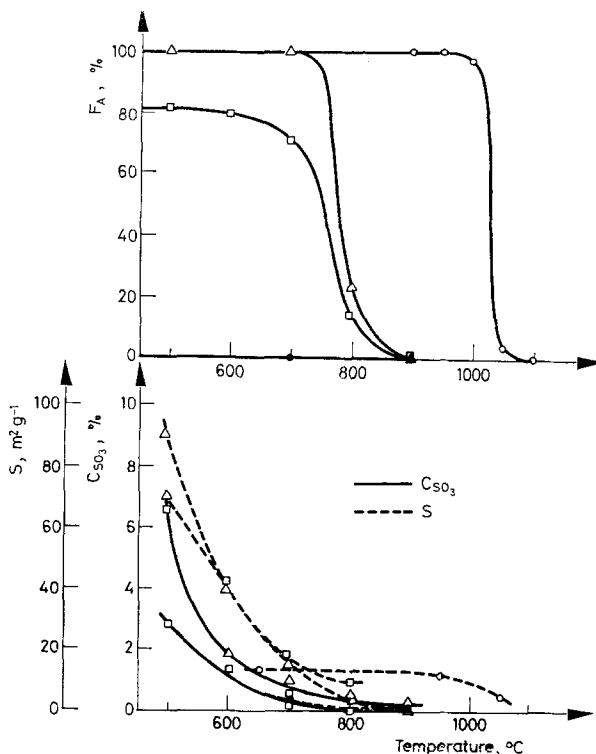


Fig. 2.  $F_A$ ,  $C_{\text{SO}_3}$  and  $S$  of hydrated  $\text{TiO}_2$  and commercial anatase as functions of the calcination temperature. Calcination time is 1 hr for  $F_A$ , and 2 hr for  $C_{\text{SO}_3}$  and  $S$ . Hydrated  $\text{TiO}_2$ : ●,  $S-2'$ ; △,  $S-3'$ ; □,  $S-5'$ ; ○, commercial anatase (C-A)

curves, which are taken as a measure of the temperature at which the transition to rutile proceeds with the greatest violence, were  $1025^\circ$  in C-A,  $775^\circ$  in  $S-3'$  and  $765^\circ$  in  $S-5'$ . The transitions in  $S-3'$  and  $5'$  seem to start as  $C_{\text{SO}_3}$  becomes less than about 1%.

#### DTA of $\text{CaSO}_4$ - $\text{TiO}_2$ and $\text{CaCO}_3$ - $\text{TiO}_2$ systems

Figure 3(A) shows the DTA curves for  $\text{CaSO}_4$  alone (a) and for  $\text{CaSO}_4$ - $\text{TiO}_2$  systems (b and c) in flowing  $\text{N}_2$ . X-ray analysis and chemical analysis of samples at the various temperatures shown by arrows on the DTA curves were carried

out to study the reaction mechanisms involved. The outlet gas was analyzed qualitatively to determine the onset temperature of decomposition of  $\text{CaSO}_4$ .

Curve (a) shows that an endothermic deflection of the DTA curve begins at about  $1100^\circ$ , followed by a sharp endothermic peak at  $1220^\circ$ . X-ray analysis of the sample taken at  $1300^\circ$  showed the presence of  $\text{CaO}$  and  $\text{CaSO}_4$ , which suggests that the endothermic deflection starting at  $1100^\circ$  is due to thermal decomposition

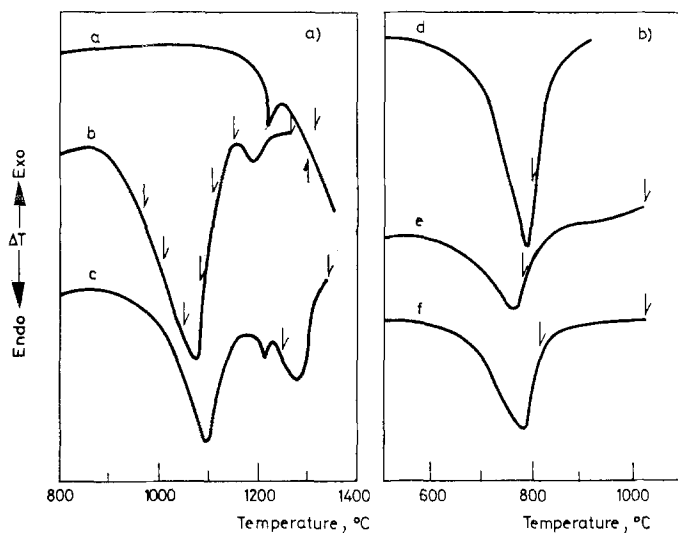


Fig. 3. DTA curves for  $\text{CaSO}_4$ - $\text{TiO}_2$  systems (A), and for  $\text{CaCO}_3$ - $\text{TiO}_2$  systems (B). Heating rate:  $5.8^\circ/\text{min}$ ; atmosphere:  $\text{N}_2$  flow (100 ml/min). Systems: (a)  $\text{CaSO}_4$  alone; (b)  $\text{CaSO}_4$  - (C-A); (c)  $\text{CaSO}_4$ -(C- $\text{R}_1$ ); (d)  $\text{CaCO}_3$  alone; (e)  $\text{CaCO}_3$ -(C-A); (f)  $\text{CaCO}_3$ -(C- $\text{R}_1$ )

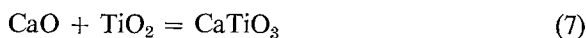
of  $\text{CaSO}_4$ . The sharp peak at  $1220^\circ$  is due to the  $\text{II} \rightleftharpoons \text{I}$  transition of remaining  $\text{CaSO}_4$ . No endothermic or exothermic peak was observed below  $1000^\circ$ .

(b) is the DTA curve for the  $\text{CaSO}_4$ - $\text{TiO}_2$  (anatase) system. An endothermic deflection begins at  $880^\circ$ , followed by two endothermic peaks, at  $1070$  and  $1190^\circ$ . Evolution of  $\text{SO}_2$  and  $\text{SO}_3$  was observed at about  $500^\circ$ . The large endothermic peak at  $1070^\circ$  corresponds to reaction (4). Some  $\text{CaSO}_4$  remains at  $1110^\circ$  and the reaction was completed at  $1265^\circ$ , but  $\text{CaO}$  was absent over the entire temperature range. On the basis of these results, it is thought that reaction (4) proceeds in two steps, at  $1070$  and  $1190^\circ$ .

(c) is the curve for the  $\text{CaSO}_4$ - $\text{TiO}_2$  (rutile) system. As in the case of (b), it is thought that reaction (4) proceeds in two steps, at  $1100$  and  $1280^\circ$ . An endothermic peak at  $1210^\circ$  corresponds to the transition of residual  $\text{CaSO}_4$ . The reason why reaction (4) proceeds stepwise may be that the rapid reaction occurs at the point of contact between two particles in the initial step, followed by the second step, which is possibly a diffusion-controlled step through the product layer. The differ-

ences in peak temperature and size for C-A and C-R<sub>1</sub> are probably due to the differences in both the structure and particle size of  $\text{TiO}_2$ .

These results were compared with those in the  $\text{CaCO}_3$ - $\text{TiO}_2$  systems. Figure 3(B) shows the DTA curves for  $\text{CaCO}_3$  alone (d) and  $\text{CaCO}_3$ - $\text{TiO}_2$  systems (e and f) in flowing  $\text{N}_2$ . (d) shows that an endothermic deflection begins at  $550^\circ$ , with a peak temperature of  $780^\circ$ . This peak corresponds to the decomposition of  $\text{CaCO}_3$  to  $\text{CaO}$ , as  $\text{CaO}$  was identified in the X-ray diffraction pattern of the sample taken at  $800^\circ$ . (e) and (f) are the curves for  $\text{CaCO}_3$ - $\text{TiO}_2$  (anatase) and  $\text{CaCO}_3$ - $\text{TiO}_2$  (rutile), respectively. There is no distinct difference between  $\text{CaCO}_3$  alone and the  $\text{CaCO}_3$ - $\text{TiO}_2$  systems, as opposed to the  $\text{CaSO}_4$ - $\text{TiO}_2$  systems. After completion of the DTA experiments, X-ray analysis of the samples showed the presence of  $\text{CaTiO}_3$  and  $\text{CaO}$ . On the basis of these results, it is considered that the reactions in the  $\text{CaCO}_3$ - $\text{TiO}_2$  systems proceed in two steps:



In this reaction system, no distinct difference in reactivity between anatase and rutile was recognized.

#### TG of $\text{CaSO}_4$ - $\text{TiO}_2$ system

On the basis of the discussions in the preceding section, it is understandable that the investigation in the  $\text{TiO}_2$ - $\text{CaSO}_4$  systems make it possible to examine the reactivity of  $\text{TiO}_2$  with different preparation histories.

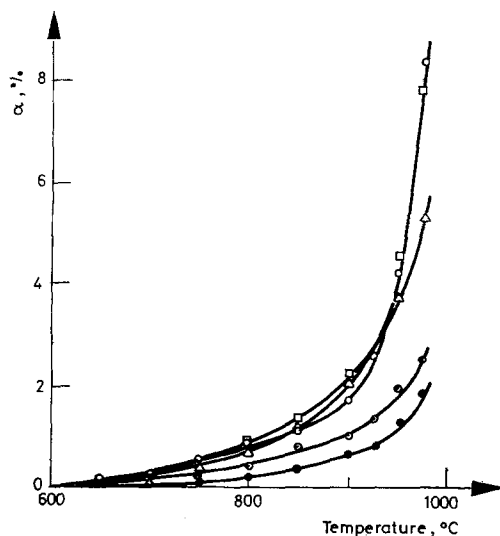


Fig. 4. TG curves for  $\text{CaSO}_4$ - $\text{TiO}_2$  systems. Heating rate:  $5^\circ/\text{min}$ ; atmosphere:  $\text{N}_2$  static (150 torr).  $\text{TiO}_2$ :  $\circ$ , C-A;  $\odot$ , C-R<sub>2</sub>;  $\bullet$ , S-2;  $\triangle$ , S-3;  $\square$ , S-5

Figure 4 shows the TG curves of the reaction between  $\text{CaSO}_4$  and five  $\text{TiO}_2$  samples. All the curves were adapted so that the weight decrease starts at  $600^\circ$ , because the weight change was negligibly small below  $600^\circ$ . The curve for the  $\text{CaSO}_4$ - $\text{TiO}_2$  (C-A) system corresponds well to the DTA curve (b) in Fig. 3. The reactivity of the anatase group (C-A, S-3 and S-5), the fractional conversion of which increases rapidly at about  $900^\circ$ , is much higher than that of the rutile group (C-R<sub>2</sub> and S-2).

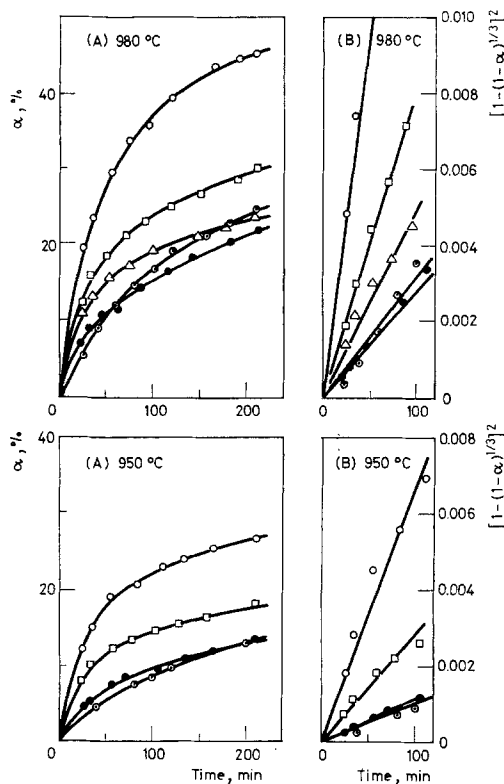


Fig. 5. (A): Isothermal TG curves at 980 (978–982°) and 950° for  $\text{CaSO}_4$ - $\text{TiO}_2$  systems in static  $\text{N}_2$  (150 torr). (B): Jander's plots of (A) in the initial reaction step.  $\text{TiO}_2$ : ○, C-A; ⊙, C-R<sub>2</sub>; ●, S-2; △, S-3; □, S-5

It is found that the reactivity of  $\text{TiO}_2$  is more dependent on the crystal form rather than the particle size ( $D$ ) or BET surface ( $S$ ), because no distinct difference in the  $D$  or  $S$  values of the samples at the preparation temperature ( $700^\circ$ ) was recognized, except for sample S-2 (preparation temperature  $900^\circ$ ). The question arises that  $S$  or  $D$  may decrease in a different manner with increasing temperature (Fig. 2), and this change influence the reactivity of  $\text{TiO}_2$  at reaction temperatures above  $700^\circ$ . However, the  $S$  values of S-2 and S-3 are approximately the same



at  $900^\circ$ , and no good correlation is obtained between the reactivity and  $S$ . On the basis of these considerations, it is thought that the formation of the intermediate thin layer,  $(\text{MO}_2 \cdot \text{CaO} \cdot \text{SO}_3)$ , assumed in reaction (2), could be affected by the crystal form of  $\text{TiO}_2$  at temperatures below around  $900^\circ$ , before the formation of  $\text{CaTiO}_3$  becomes significant. If the  $\text{TiO}_2$  samples are regarded as spherical in shape and  $0.5 \mu\text{m}$  in diameter, an intermediate thin layer of about  $300 \text{ \AA}$  is evaluated when the fractional conversion is 30%.

#### *Isothermal study in $\text{CaSO}_4\text{--TiO}_2$ system*

Figure 5(A) shows the isothermal TG curves at  $980$  ( $978\text{--}982^\circ$ ) and  $950^\circ$  for the reaction between  $\text{CaSO}_4$  and five  $\text{TiO}_2$  samples with different preparation histories. From the slopes of the  $\alpha$  vs.  $t$  curves, a great difference in reaction rate can

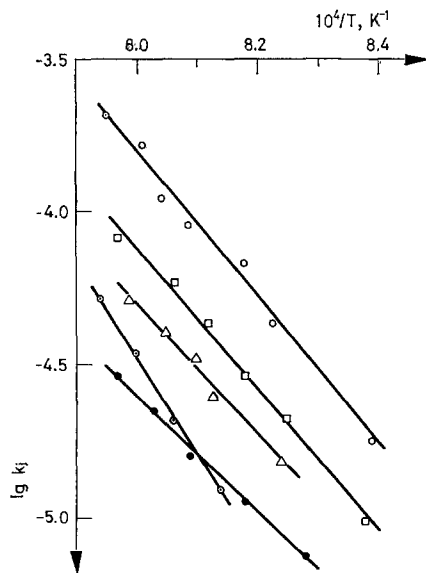


Fig. 6.  $\log k_j$  as a function of  $1/T$  for  $\text{CaSO}_4\text{--TiO}_2$  systems.  $\text{TiO}_2$ :  $\circ$ , C-A;  $\odot$ , C-R<sub>2</sub>;  $\bullet$ , S-2;  $\triangle$ , S-3;  $\square$ , S-5

be found between the  $\text{TiO}_2$  samples ( $\text{C-A} > \text{S-5} > \text{S-3} > \text{S-2} \approx \text{C-R}_2$ ), as regards the initial reaction step up to about 60 min, while no distinct difference is recognized after 100 min.

Figure 5(B) shows the plots of the data for the initial reaction step as a function of time, according to Jander's equation,  $k_j t = (1/1-\alpha^{1/3})^2$ , which is based on diffusion. Kinetic data,  $\log k_j$  as a function of  $1/T$ , are shown in Fig. 6 for five  $\text{CaSO}_4\text{--TiO}_2$  systems. Activation energies estimated on the basis of Arrhenius' equation were 452, 527, 368, 389 and 435 kJoule/mole for C-A, C-R<sub>2</sub>, S-2, S-3

and S-5, respectively. No interesting information in connection with the reactivity of  $\text{TiO}_2$  was obtained from the activation energy values estimated.

Figure 7 shows the relations between  $\log k_i$  obtained in the initial reaction step as a measure of the reactivity of  $\text{TiO}_2$  (in Fig. 6) and the transition temperatures

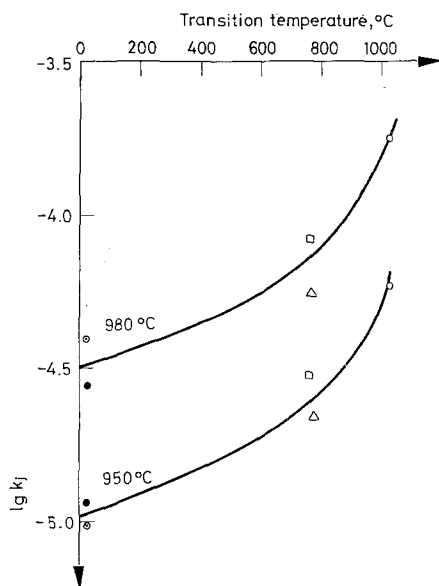


Fig. 7.  $\log k_i$  at  $980^\circ$  and  $950^\circ$  for  $\text{CaSO}_4$ - $\text{TiO}_2$  systems vs. transition temperatures of  $\text{TiO}_2$  samples. The symbols are the same as in Fig. 6

of anatase to rutile (inflection points of the curves in Fig. 2) at the reaction temperatures of  $950$  and  $980^\circ$ . It is seen that the greater the stability of the anatase form up to higher temperature, the higher the reactivity of  $\text{TiO}_2$ .

### Conclusions

(1) In order to estimate the reactivities of solids, it is necessary to note the behaviors in the initial reaction step in the selected reaction system, because the reactivity is occasionally governed by the initial reaction step [5] occurring prior to the main reaction generally used for the reactivity measurements of solids.

(2) In the present study of the initial reaction steps of  $\text{CaSO}_4$ - $\text{TiO}_2$  systems, it was thought that the reactivity of the  $\text{TiO}_2$  is dependent on the structure difference due to the different preparation histories.

### References

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RÉSUMÉ — On a étudié par TG, ATD et expériences cinétiques isothermes l'influence du mode de préparation de  $\text{TiO}_2$  sur sa réactivité lors de la réaction en phase solide des systèmes  $\text{CaSO}_4$  —  $\text{TiO}_2$ .

Afin d'estimer la réactivité des solides, il a été nécessaire d'observer leurs comportements lors de l'étape initiale de la réaction dans le système réactionnel choisi. On en a conclu que la réactivité de  $\text{TiO}_2$  dépend des différences de structure dues aux différents modes de préparation et que ce composé est d'autant plus stable, à températures élevées, sous la forme d'anatase, que la réactivité de  $\text{TiO}_2$  est forte.

ZUSAMMENFASSUNG — Der Einfluß der Herstellungsgeschichten von  $\text{TiO}_2$  auf seine Reaktivität bei der Festphasenreaktion im System  $\text{CaSO}_4$  —  $\text{TiO}_2$  wurde durch TG, DTA und isotherme kinetische Versuche untersucht.

Zur Abschätzung der Reaktivität der Festkörper mußte ihr Verhalten während des Anfangsschrittes der Reaktion im gewählten System beobachtet werden. Es wurde gefolgert, daß die Reaktivität des  $\text{TiO}_2$  von den durch verschiedene Herstellungsgeschichten verursachten Strukturunterschieden abhängt und bis zu höheren Temperaturen umso stabiler in der Anatase-Form ist, je höher die Reaktivität des  $\text{TiO}_2$  ist.

Резюме — С помощью ТГ, ДТА и изотермических кинетических методов было изучено влияние способов приготовления  $\text{TiO}_2$  на его реакционную способность в твердотельной реакции в системе  $\text{CaSO}_4$ - $\text{TiO}_2$ . Для установления реакционной способности твердых образцов, было исследовано их термическое поведение в начальной стадии реакции в избранной реакционной системе. Сделано заключение, что реакционная способность  $\text{TiO}_2$  зависит от его структурных особенностей, обусловленных различными способами приготовления и наиболее устойчива в форме анатаза к высоким температурам и где более высокая реакционная способность  $\text{TiO}_2$ .